Multi-Scale Unit-Cell Analysis of Textile Composites' Finite Deformation Stiffness Characteristics

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<u>Overview</u>

1. Objectives: Material models for textile composites to facilitate structural analysis

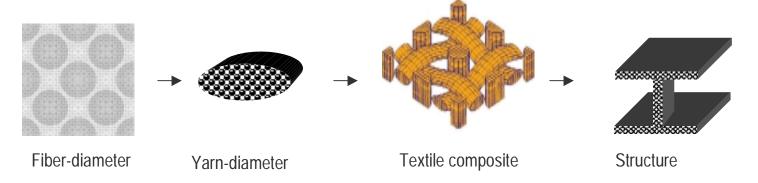
Enhanced understanding of textile composites' property-structure relations

- 2. Approach: Repeated unit-cell analysis and homogenization on multiple scales.
- 3. Results on fiber-diameter scale
- 4. Results on textile unit-cell scale



Research Objectives

- Identify suitable forms of constitutive models for textile composites
- Evaluate suitability and consistency of model forms using unit-cell analysis
- Better understand property-structure relations
 - specific interest: geometric stiffness effects
 - positive tension stiffening / negative compression stiffening
- Length scales considered





Essence of Unit-Cell Homogenization (for heterogeneous, periodic media)

- On a given length scale at which the material is heterogeneous (micro scale), apply an average stress or average deformation to a detailed model (unit cell)
- For each loading, compute detailed, equilibrium microscale stress and deformations fields.
- Take the spatial average of the "microscale" stress and deformation fields, to get their "macroscopic" correspondent.
- Develop/calibrate a constitutive model that adequately relates the macroscale stresses and deformations.
- When performing analysis of the system on the "macroscale" use the "homogenized" constitutive model to represent the medium.



Micro-/Macro-scale Notation

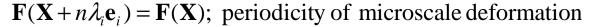
- Periodic medium and unit cell
- Microscale stress/deformation

$$\sigma(\mathbf{X}) = \Sigma + \sigma^*(\mathbf{X});$$

$$\mathbf{F}(\mathbf{X}) = \mathbf{\Phi} + \mathbf{F}^*(\mathbf{X});$$

$$<\sigma^*(\mathbf{X})>=0;$$

$$<\mathbf{F}^{*}(\mathbf{X})>=\mathbf{0};$$



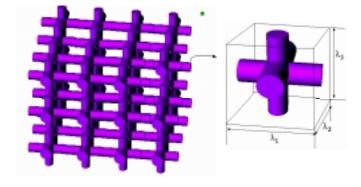
$$\sigma(\mathbf{X} + n\lambda_i \mathbf{e}_i) = \sigma(\mathbf{X})$$
; periodicity of microscale stress

• Averaging stress/deformation to find macroscale correspondents

$$\Sigma = \langle \mathbf{\sigma} \rangle = \frac{1}{V} \int_{\Omega_s} \mathbf{\sigma} \, d\Omega_s;$$

$$\mathbf{\Phi} = \langle \mathbf{F} \rangle = \frac{1}{V} \int_{\Omega_s} \mathbf{F} \, d\Omega_s;$$





PROCESS: Deformation-Controlled Loading of Unit-Cell

- Specify an average state of deformation Φ for the unit cell.
- Apply a consistent "homogeneous" displacement field $\mathbf{u} = \Phi \cdot \mathbf{X}$ to unit cell.
- To achieve stress-field equilibrium on microscale, solve for the additive, periodic, heterogeneous displacement field u*(X).
- Resulting equilibrium displacement field: $\mathbf{u}(\mathbf{X}) = \Phi \cdot \mathbf{X} + \mathbf{u}^*(\mathbf{X})$
- For each macroscopic state of deformation Φ , compute the corresponding macroscopic state of stress Σ .
- Consider the Σ versus Φ behavior of the unit cell model.
- Provide and calibrate a macro-scale constitutive model $\Sigma = \Sigma(\Phi)$.



Symmetric, Conjugate, Macro Stress/Strain Measures

- Using conjugate macroscopic stress/strain measures ensures energy conservation between micro- and macro-scales.
- Nemat-Nassar (2000) demonstated/used conjugacy between macroscale deformation gradient Φ and the macroscale nominal stress <**P**>.

$$\langle \mathbf{P} : \dot{\mathbf{F}} \rangle = \langle \mathbf{P} \rangle : \langle \dot{\mathbf{\Phi}} \rangle$$

• It is preferred to develop constitutive models in terms of symmetric, macroscopic stress and deformation measures. Here, we use:

$$\hat{\mathbf{\Sigma}} = \langle \mathbf{P} \rangle \mathbf{\Phi}^{-T};$$

$$\hat{\mathbf{E}} = \frac{1}{2} [\mathbf{\Phi}^{T} \mathbf{\Phi} - \mathbf{I}];$$

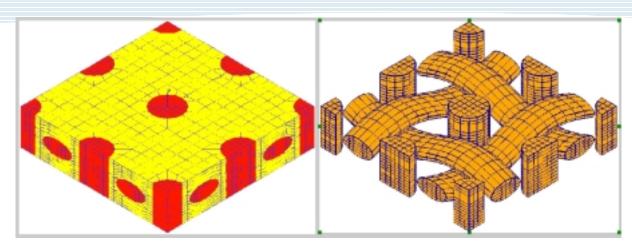
• These symmetric measures satisfy the following conjugacy relationship:

$$\hat{\Sigma} : \dot{\hat{\mathbf{E}}} = \langle \mathbf{P} : \dot{\mathbf{F}} \rangle = \langle \mathbf{S} : \dot{\mathbf{E}} \rangle$$



Major Challenge

- Constructing appropriate unit cell models that capture:
 - material structure
 - material scale stress/strain fields
- Special issues pertaining to composites:
 - complex material interfaces
 - displacement periodicity (requires external face matching)
 - two-sidedness: meshing both sides of each material interface

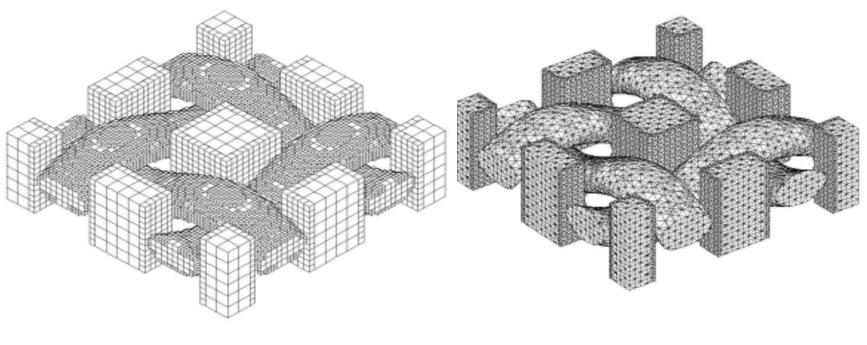


(a) unit-cell model for textile composite

(b) woven yarn reinforcements



Alternative Unit-cell Meshing Approaches

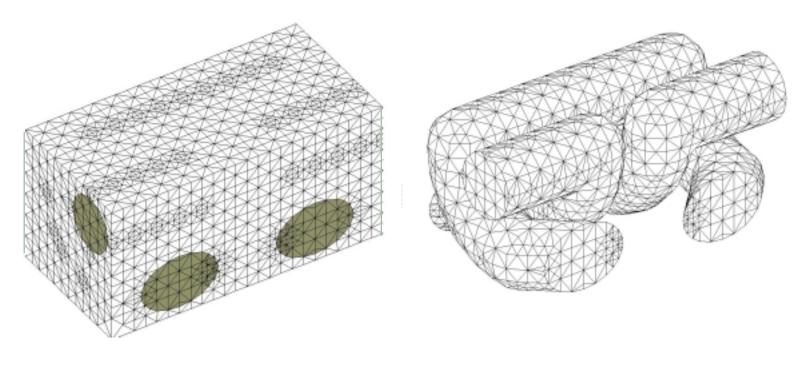


(a) voxel-based mesh

(b) tetrahedral mesh



Unit Cell Model for Knitted Composite

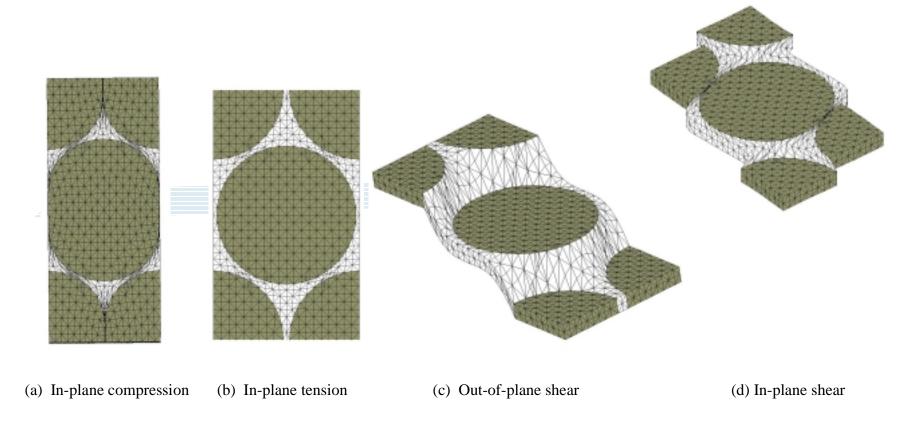


(a) exterior of unit cell

(b) reinforcements only



<u>Unit-Cell Analysis at</u> <u>Fiber-Diameter Scale</u>





Transversely Isotropic Hyperelastic Model for Aligned Fiber Materials

Model by Bonet and Burton (1998)

$$\psi = \psi(I_1, I_2, I_3, I_4, I_5) = \psi_{inh} + \psi_{ti}$$

$$\psi_{inh} = \frac{1}{2}\mu(I_1 - 3) - \mu \ln J + \frac{1}{2}\lambda(J - 1)^2$$

$$\psi_{ti} = [\alpha + \beta \ln J + \gamma(I_4 - 1)](I_4 - 1) - \frac{1}{2}\alpha(I_5 - 1)$$

where

$$I_1 = tr(\mathbf{C}); \quad I_2 = \mathbf{C} : \mathbf{C}; \quad I_3 = \det(\mathbf{C}) = \mathbf{J}^2;$$

 $I_4 = \mathbf{A} \cdot \mathbf{C} \cdot \mathbf{A}; \quad I_5 = (\mathbf{A} \cdot \mathbf{C}) \cdot (\mathbf{C} \cdot \mathbf{A})$

and ${\bf A}$ is axis of symmetry or fiber director in undeformed configuration



Coefficient Estimation for Aligned-Fiber Composite

- Coefficient for Bonet and Burton's model $\alpha = (\lambda, \mu, \alpha, \beta, \gamma)$
- Under Defomation-controlled homogenization

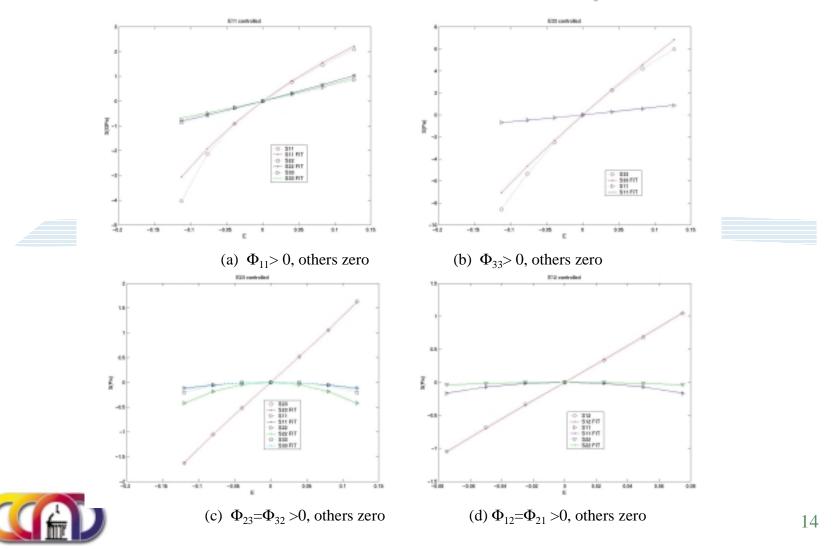
$$\min_{\boldsymbol{\alpha}} \sum_{k} \int_{t_1}^{t_2} \left\| \frac{\hat{\boldsymbol{S}}^{k}(\tau) - \tilde{\boldsymbol{S}}^{k}(\boldsymbol{\alpha}, \tau)}{\hat{\boldsymbol{S}}^{k}(\tau)} \right\| d\tau$$

where

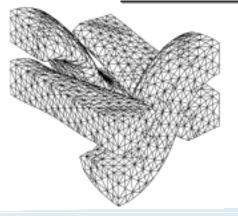
 $\hat{\mathbf{S}}^{k}(\tau)$ is homogenized stress under k_{th} strain-controlled case $\tilde{\mathbf{S}}^{k}(\boldsymbol{\alpha},\tau)$ is second PK stress from proposed model.

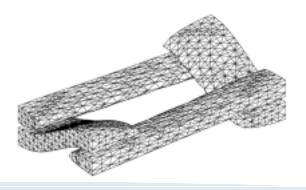


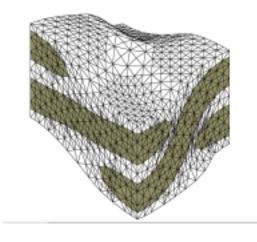
Homogenized, Macro-scale Stress-Strain Behavior of Yarn Composite

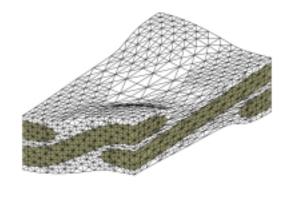


Analysis at the Textile Scale for Plain-Weaves







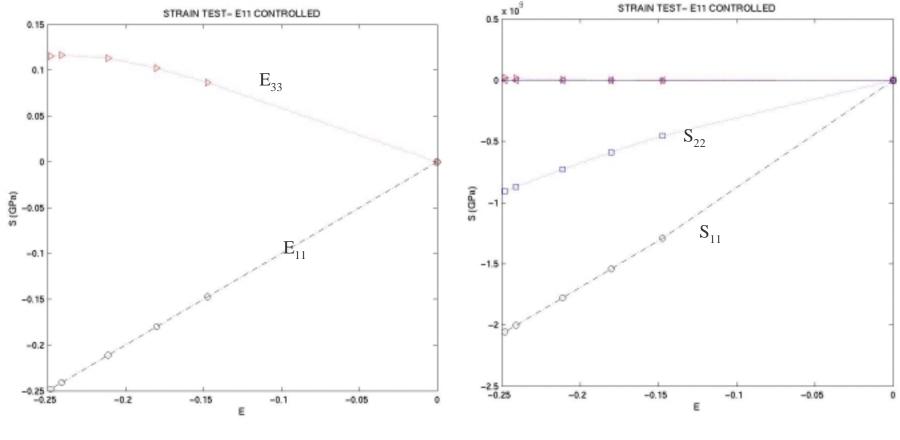


(a) uni-axial compression

(b) uni-axial tension



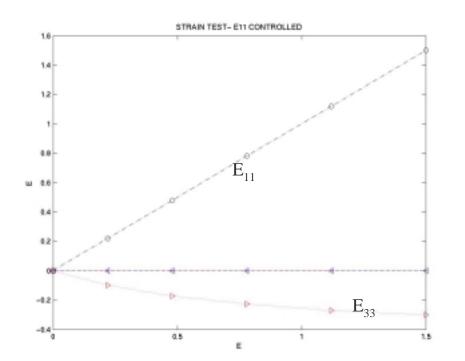
Stress-Strain Behavior of Plain-weave Textile Composite

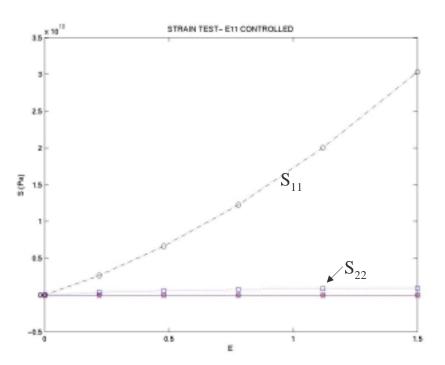




(a) uni-axial compression

Stress-Strain Behavior of Plain-weave Textile Composite (Continued)





(b) uni-axial tension



SUMMARY

- Have considered finite-deformation unit cell analysis of textile composites on multiple length scales.
- General Contributions:
 - Development of automatic mesh generator suitable for most periodic composites.
 - · Problem formulation at finite deformations
 - Symmetric, conjugate stress and strain measures
- Hyperelastic analysis of plain-weave composite
 - · finite deformation at fiber diameter scale
 - material modeling of yarn composite
 - tendency for compression buckling and tension stiffening observed



Extensions of Current Work

- Interface modeling with experimental testing program
 - For example, if graded material properties at fiber-matrix interface are measured, incorporate these into models to capture their effect.
- Material Failure Considerations (Fiber-diameter scale):
 - incorporate and model effects of fiber breakage & matrix cracking
- Material Modeling at Textile Scale
 - suitable orthotropic hyperelastic model that captures compression buckling and tension stiffening effects
 - damage model to capture gross effects of fiber-diameter scale material failure

